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Homogénéisation numérique du comportement de matériaux hétérogènes

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CEA Saclay DEN/DMN/SRMA



1 – Homogénéisation numérique

1.1 : Principes

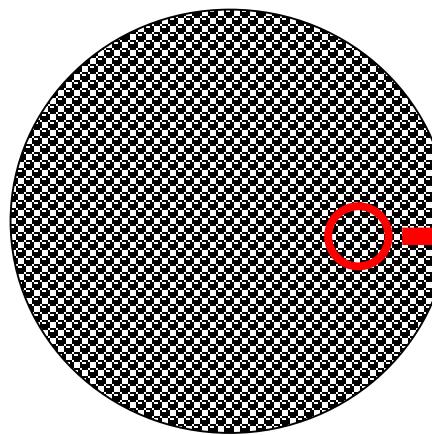
1.2 : Procédures CAST3M

2 – Applications

2.1 : Effet de la macro-porosité sur le comportement élastique des composites SiC/SiC

2.2 : Effets des joints de grains sur la conductivité thermique des polycristaux de SiC

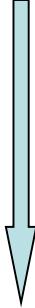
Volume Elémentaire
Représentatif



Comportement Effectif



Insensitive au choix des C.L.



Grosses « machines »!

Volume Elémentaire
Statistique

Comportement Apparent

Sensible au choix des C.L.



Moyenne statistique

Estimations du
comportement Effectif
(Petites « machines »!)



Homogénéisation numérique : principes

Propriétés en élasticité

$$\overline{K}_{CL}^{app}$$

Moyenne statistique sur un grand nombre de VES

$$\overline{K}_P^{app}$$

est une **estimation** de K^{eff}

$$\overline{K}_{DH}^{app}$$

est une **borne supérieure** de K^{eff}

$$\left((K_{CH}^{app})^{-1} \right)^{-1}$$

est une **borne inférieure** de K^{eff}

taille des VES



Écart entre les bornes

Vitesse de convergence statistique

Durée de chaque calcul



Compromis temps/précision



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Homogénéisation numérique : procédures CAST3M

➤ Application de conditions aux limites

RIGCL0 DEPI0 = NOMPROC MAIL0 TAB1;

* TAB1 = tenseur imposé

* Utilisation : DEP0 = RESOU (RIGMAT0 ET RIGCL0) DEPI0;

MECANIQUE

CLPD	Périodique	Déformation imposée
CLPC	Périodique	Contrainte imposée
CLCH	Contrainte homogène	Contrainte imposée
CLDH	Déformation homogène	Déformation imposée
CLDHC	Déformation homogène	Contrainte imposée
CLMI1	Mixte 1	Contrainte imposée
CLMI2	Mixte 2	Contrainte imposée

THERMIQUE

CLTH	Gradient de T homogène	Gradient de T
CLPT	Périodique	Gradient de T
CLFH	Flux homogène	Flux

➤ Evaluation du comportement apparent

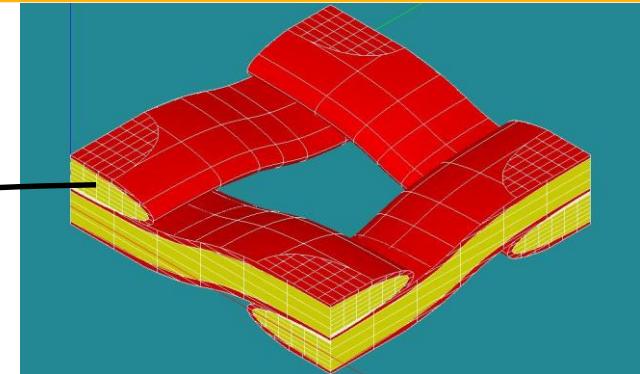
KAPP0 CONT0 DEF0 = KEFF MOD0 MAT0 NOMPROC0 AMPL0 CONV0 VISU0;
KEFFT

Homogénéisation numérique : Applications

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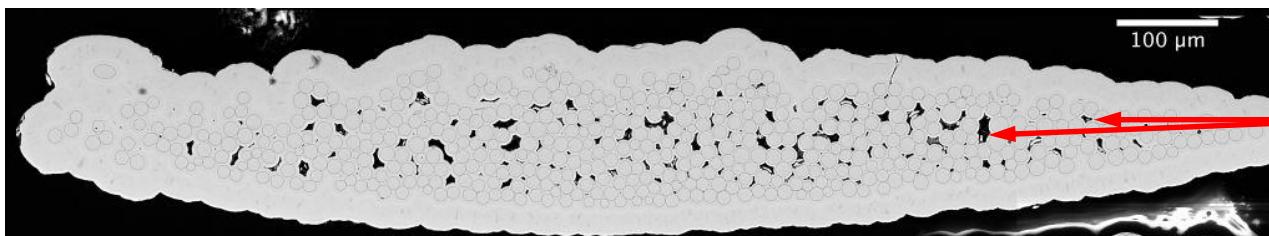
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- ✓ 2D plain weave architecture
- ✓ 13 plies
- ✓ Fibres : Hi-Nicalon S ($\sim 13\mu\text{m}$)
- ✓ Matrix : CVI



3D geometrical modelling of the plain weave pattern

SEM section of a tow:

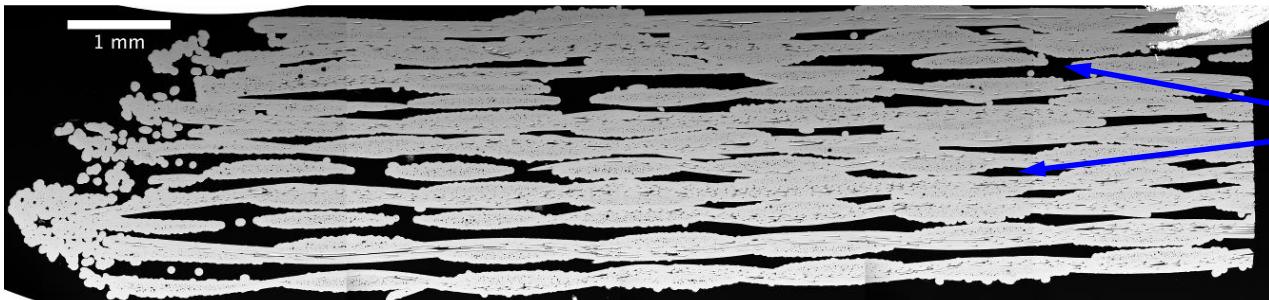


0.1mm

100 μm

micro-porosity ~5%
Fibres (500, ~55%)
(=> JNM, 2009
+ PhD thesis C. Château,)

SEM section of the whole composite



1mm

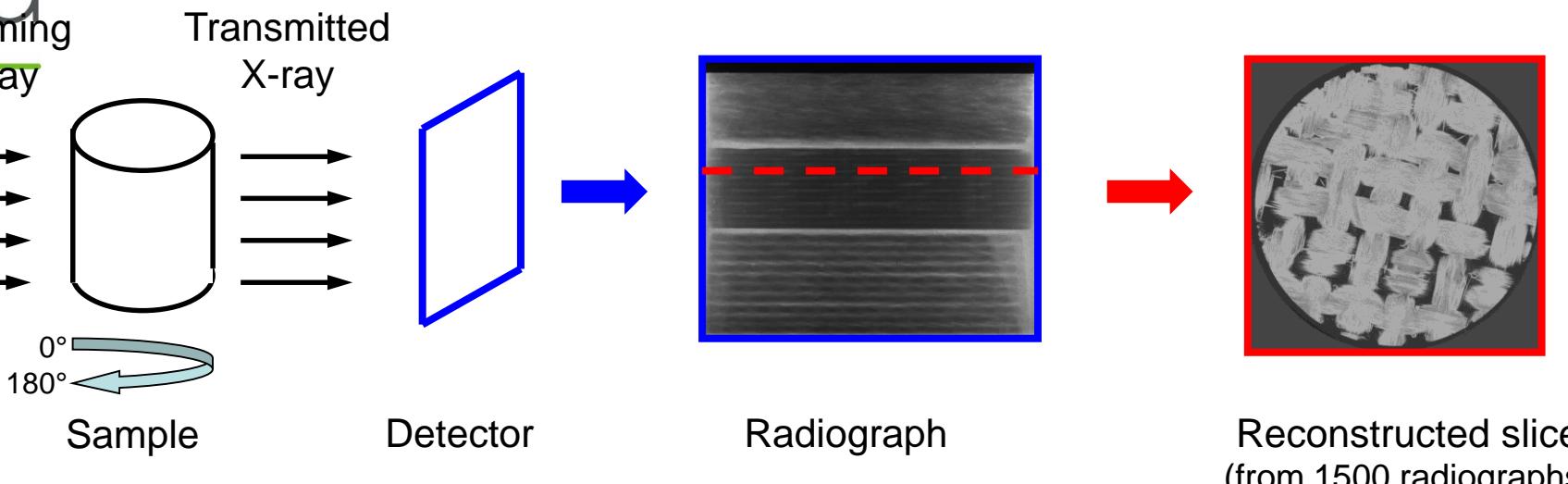
macro-porosity
3D open porosity
very complex shape

Homogénéisation numérique : Applications

➤ Principle of X-ray tomography

Incoming

X-ray



➤ Main characteristics of X-ray tomography on the ID19 beamline (ESRF)

- Resolution (« voxel » size) : 5.02³µm³
- Field of view : 10.28mm (diameter) x 8.53mm (height)

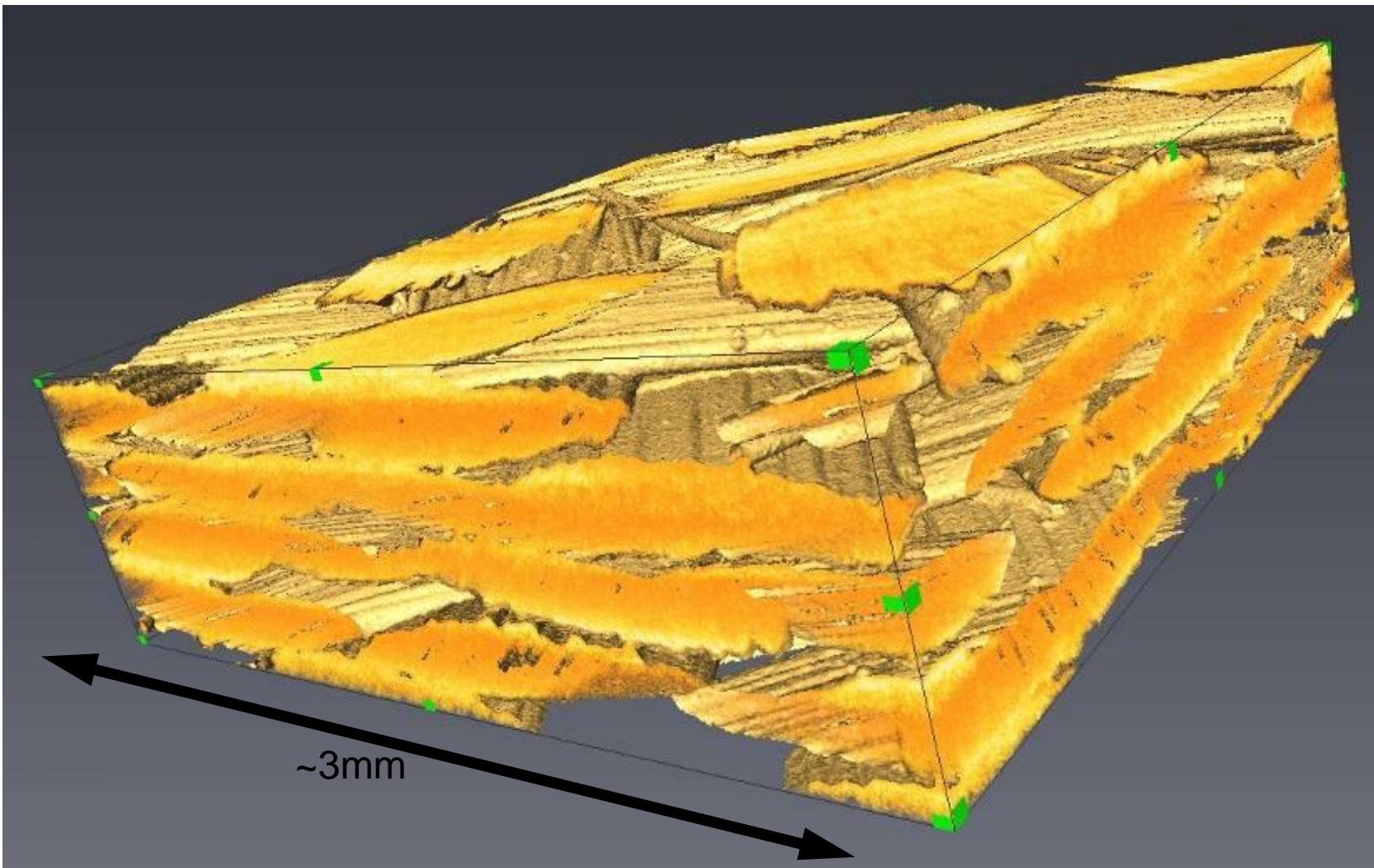
Observation of the macro-porosity on a representative volume element



Homogénéisation numérique : Applications

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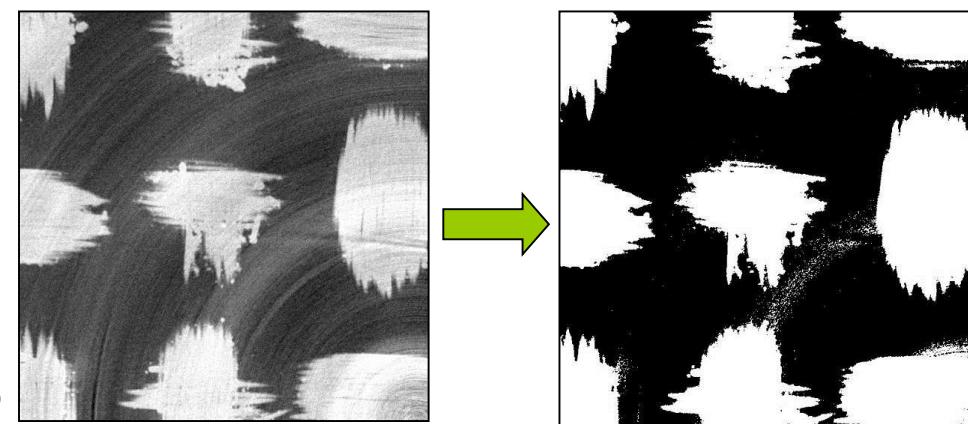
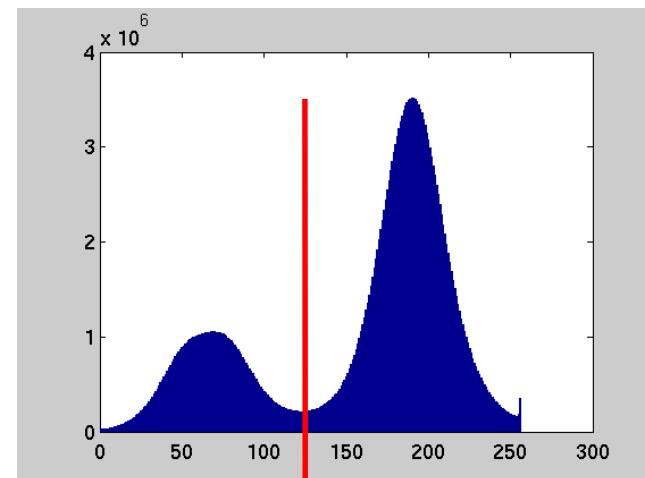
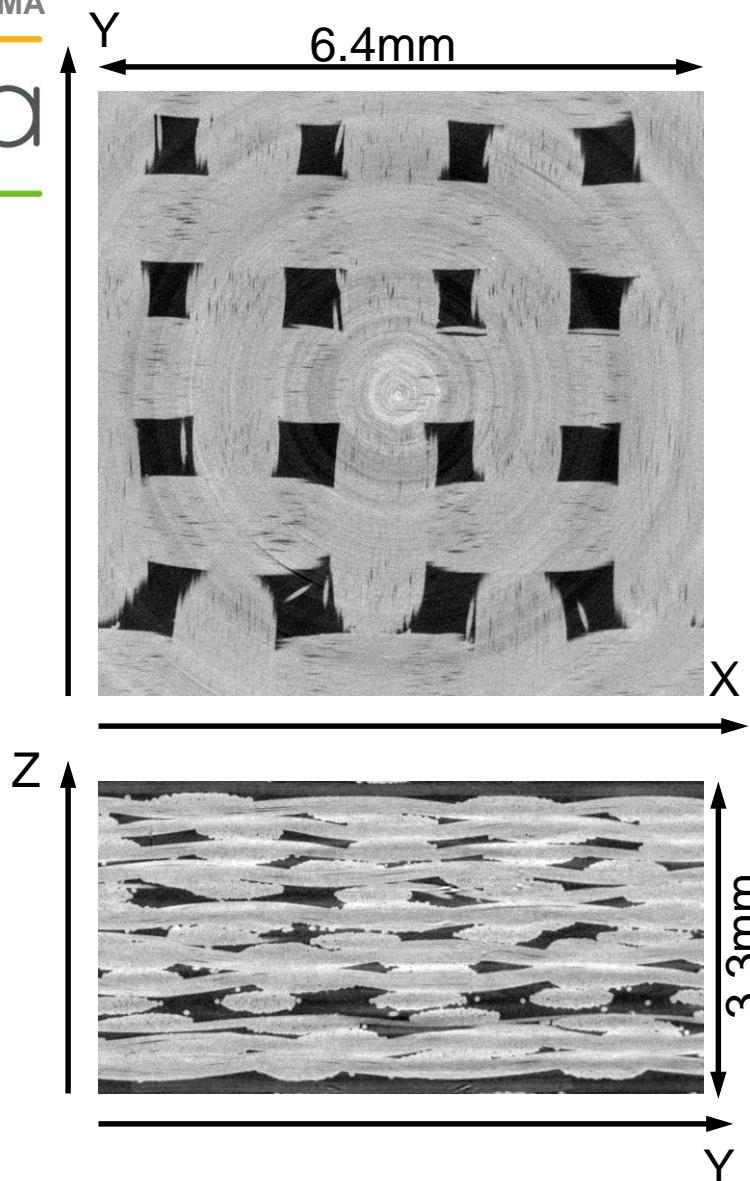
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Homogénéisation numérique : Applications

➤ Region of interest

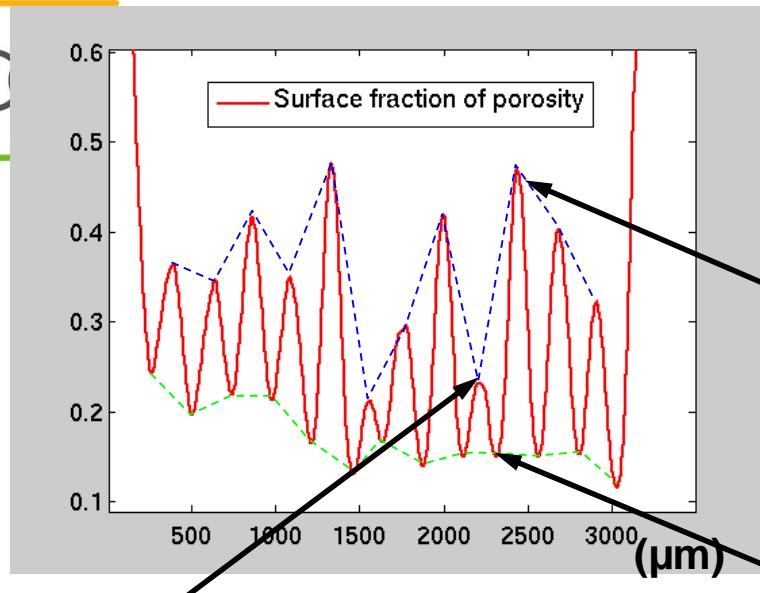
➤ Threshold operation



A red arrow points to a red-bordered box containing the formula $F_{por} = 26,5\%$.

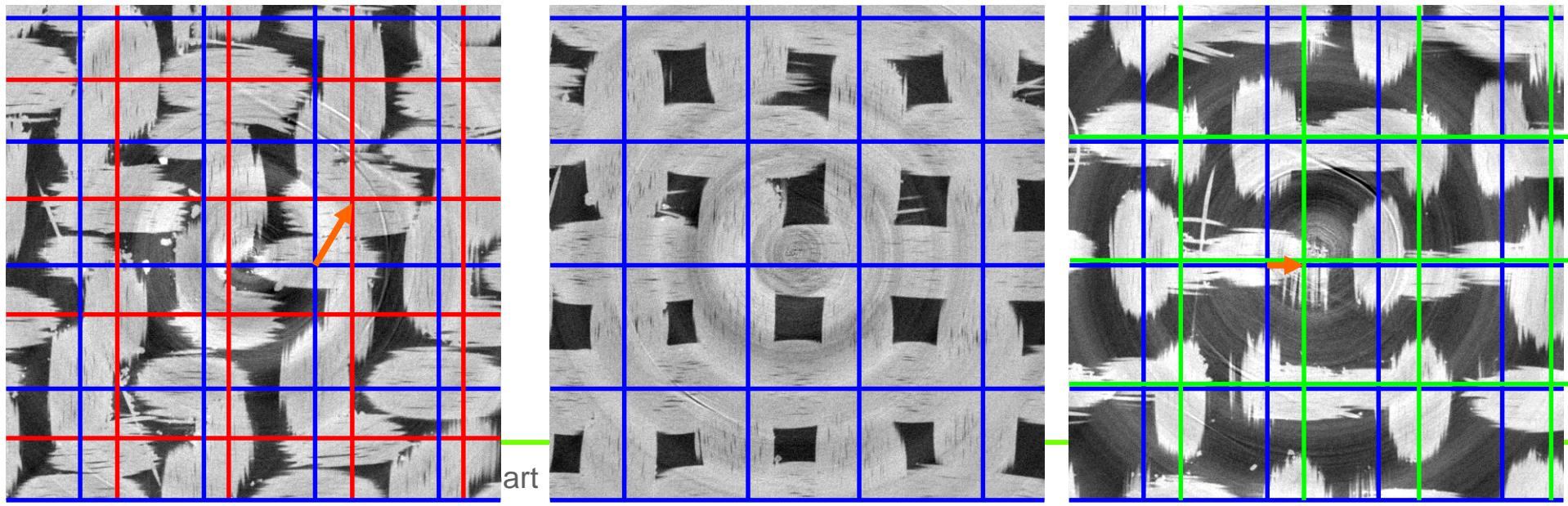
Homogénéisation numérique : Applications

- Evolution of the surface fraction of porosity in the Z-direction

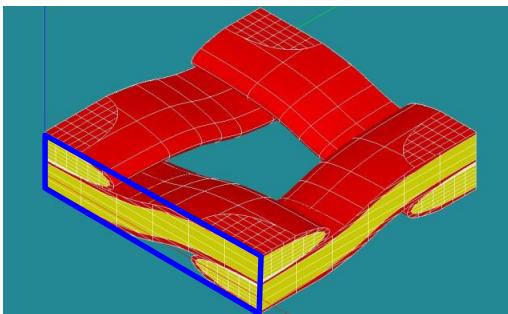


Minimum = median plane **within** a ply

Maximum = median plane **between** two plies



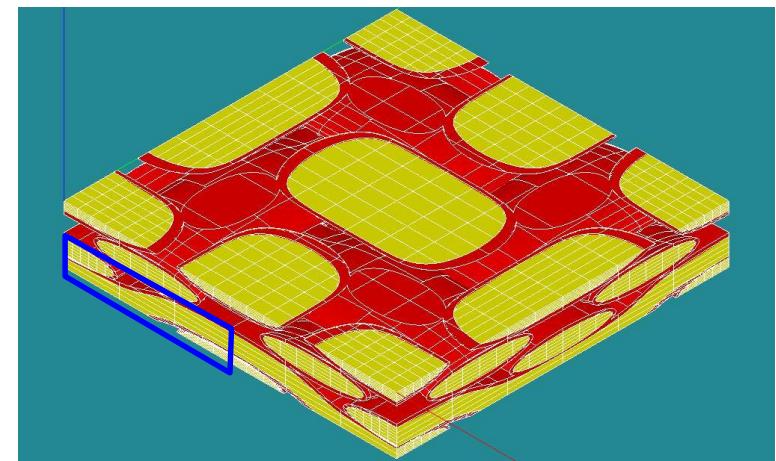
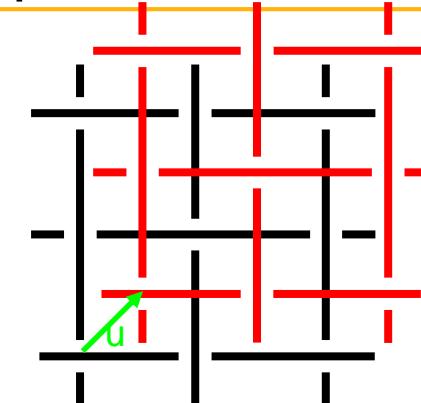
- Discussion with a geometrical model



Periodic unit-cell = one ply



Fporosity = 41%



Periodic unit-cell = two plies

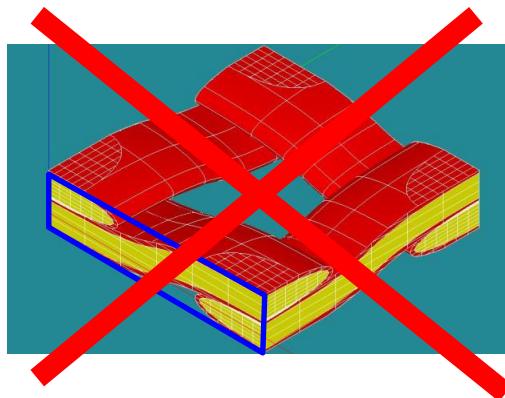


Fporosity = 21%

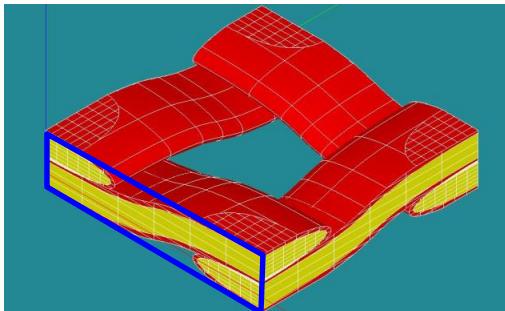
Volume fraction of porosity strongly depends on the stacking of the plies

➤ Consequences on the micro-mechanical modelling

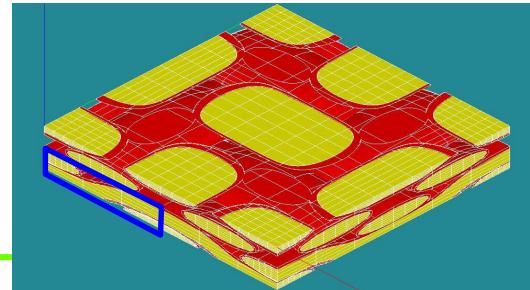
- Only one Volume Element (or unit-cell) with only one ply **is not sufficient**



- To account for the variability of the stacking of the plies :
 - **several** volume elements (« Statistical Volume Elements »)
 - **at least two plies** per volume element



+



+ ...



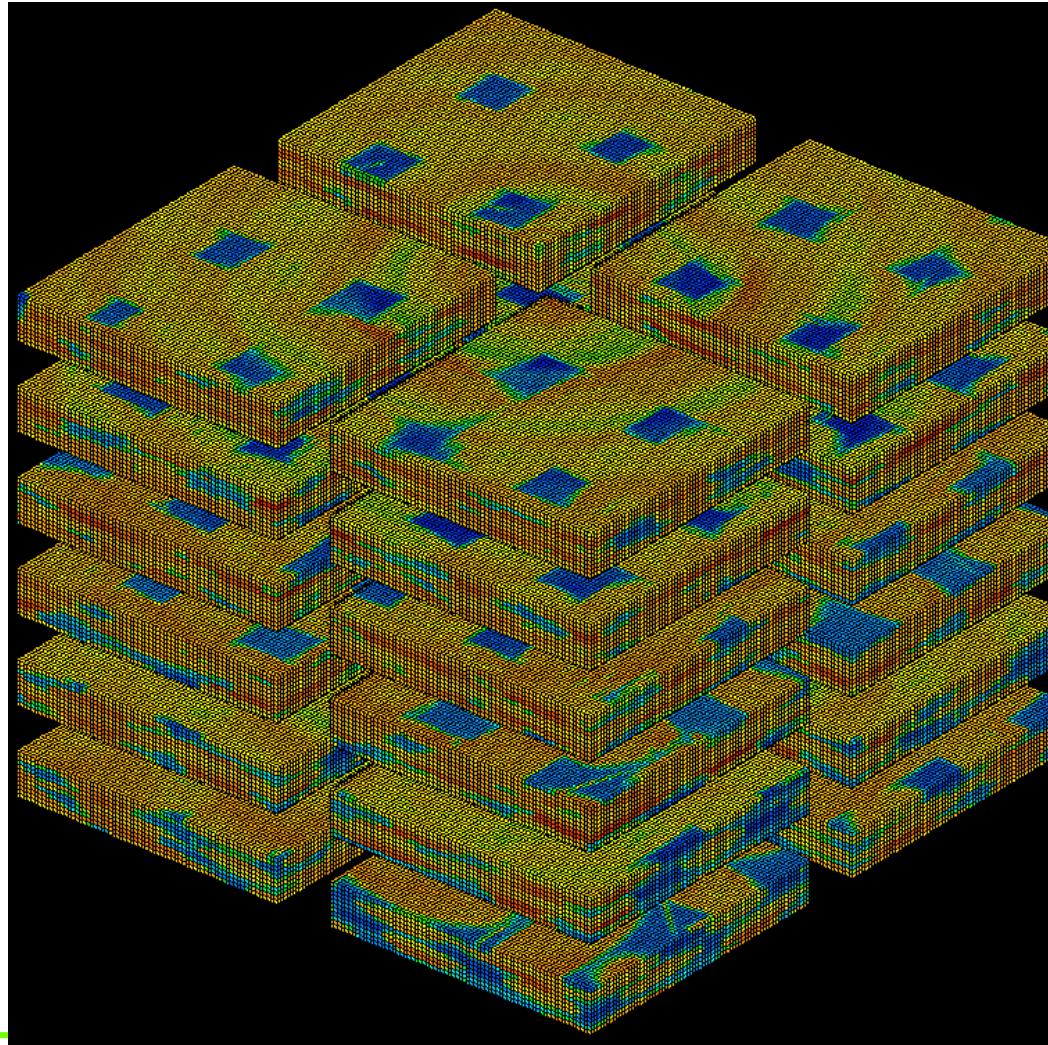
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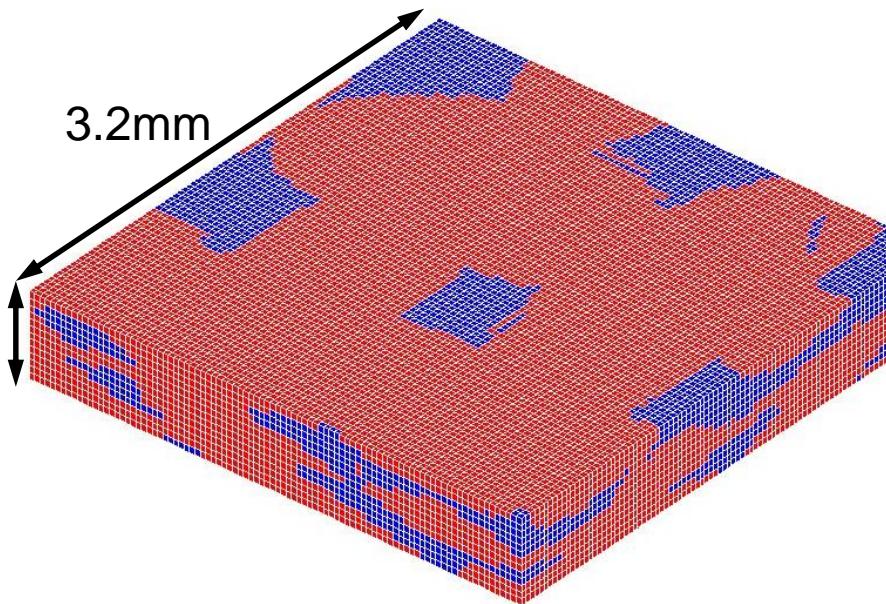
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- Definition of the « Statistical Volume Elements »

24 SVE directly extracted from the tomographic image



Homogénéisation numérique : Applications



(110000elements)

Material properties

Composite

$$E = 400 \text{ GPa}, \\ \nu = 0.3$$

Porosity

$$E = 0.0004 \text{ GPa} \\ \nu = 0.3$$

Homogénéisation numérique : Applications

➤ Results : anisotropy of the « effective » behaviour

K_{PBC}^{eff}

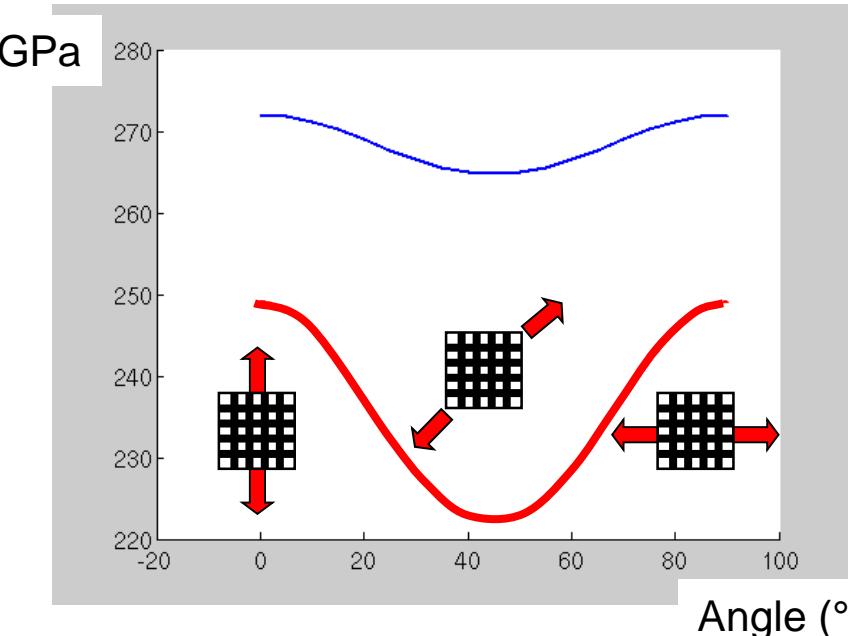
$$K_{PBC}^{eff} = \begin{pmatrix} 289 & 83 & 58 & 1 & 2 & 0 \\ & 278 & 53 & 1 & 1 & 0 \\ & & 161 & 0 & 1 & 0 \\ & & & 84 & 0 & 0 \\ & & & & 52 & 0 \\ & & & & & 39 \end{pmatrix}$$



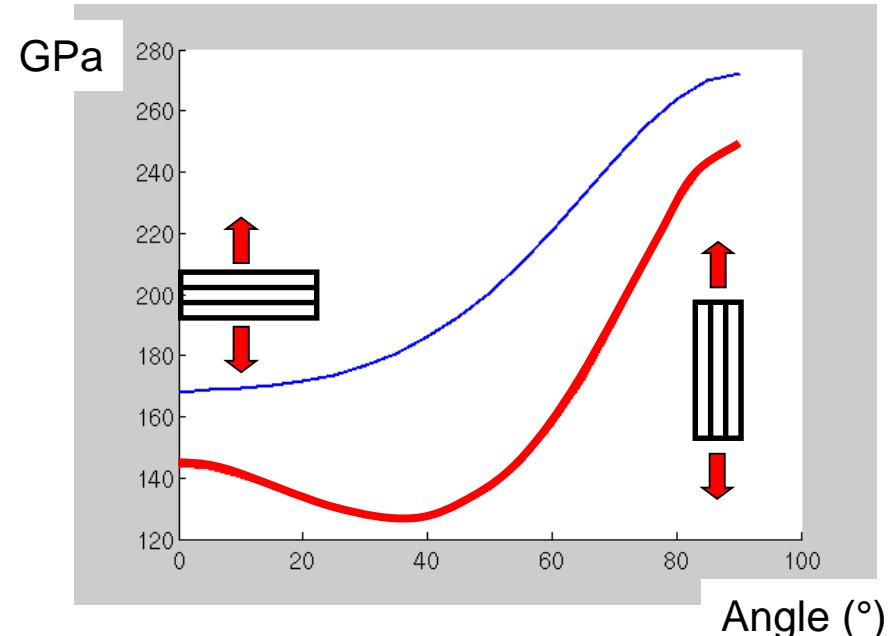
$$\begin{pmatrix} 284 & 83 & 56 & 0 & 0 & 0 \\ & 284 & 56 & 0 & 0 & 0 \\ & & 161 & 0 & 0 & 0 \\ & & & 84 & 0 & 0 \\ & & & & 46 & 0 \\ & & & & & 46 \end{pmatrix}$$

Exact quadratic symmetry

Young modulus in the (X-Y)plane



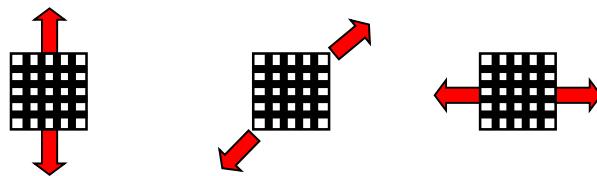
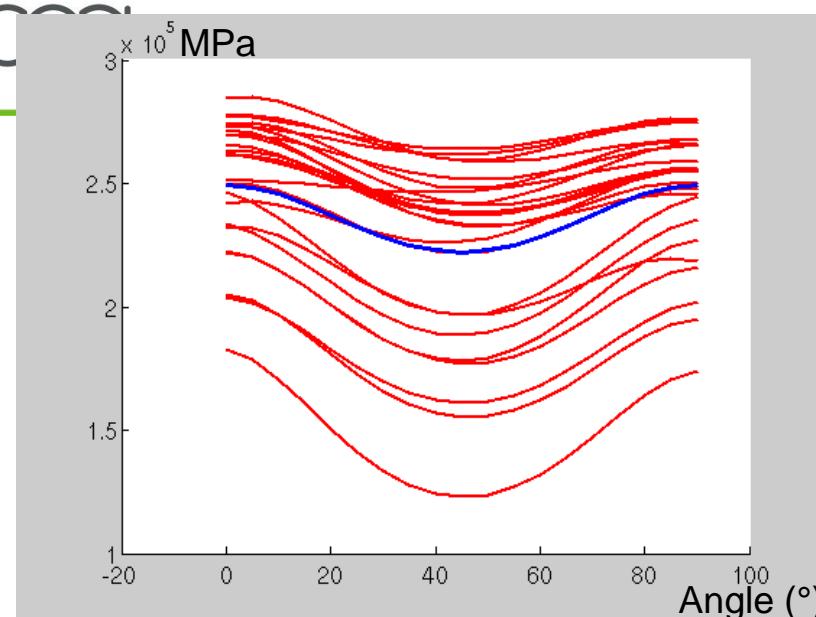
Young modulus in the (X-Z)plane



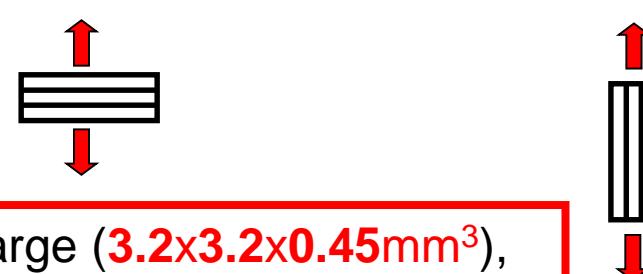
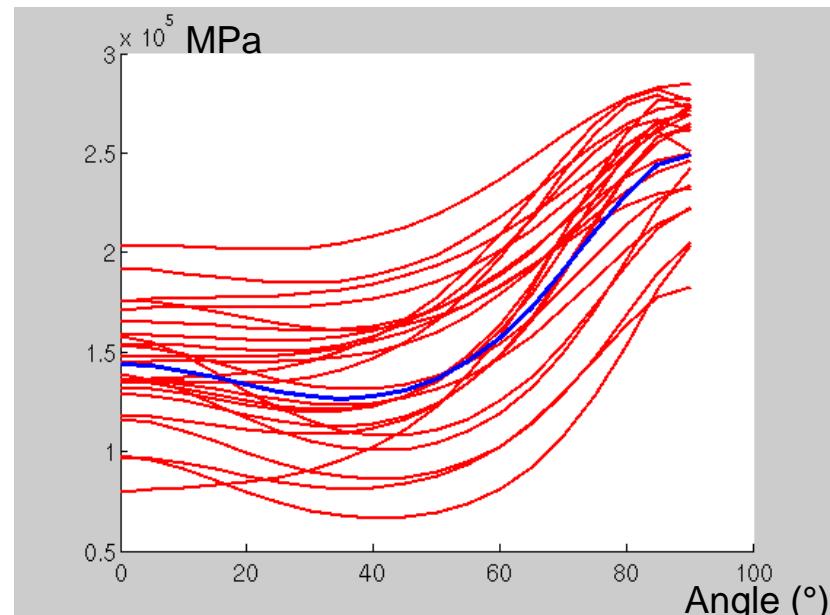
Homogénéisation numérique : Applications

- Results : fluctuations of the « apparent » behaviours

Young modulus in the (X-Y)plane



Young modulus in the (X-Z)plane



Even if the size of the volume element is quite large (**3.2x3.2x0.45mm³**),
important fluctuations are observed on the “apparent” behaviour



« Open » question : how to take into account this variability in the calculations of a SiC/SiC component?



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Homogénéisation numérique : Applications

CONCLUSIONS

2D SiC/SiC composites have been characterized by X-ray tomography
the variability of the stacking of the plies strongly influences the porosity distribution

To account for this variability in micro-mechanical modeling

- several volume elements (« Statistical Volume Element » (SVE))
- at least two plies per volume elements

The effect of the macroporosity has been evaluated on the the «elastic » behaviour:

- the macroprorosity induces an anisotropic « effective » behaviour
- the macroporosity induces important fluctuations, even for a quite important size of the volume element (3.2x3.2x0.45mm³),

FUTURE PROSPECTS

Thermal behaviour (on-going calculations, very similar results)

Introduction of the anisotropy of the tow (due to the micro-porosity)



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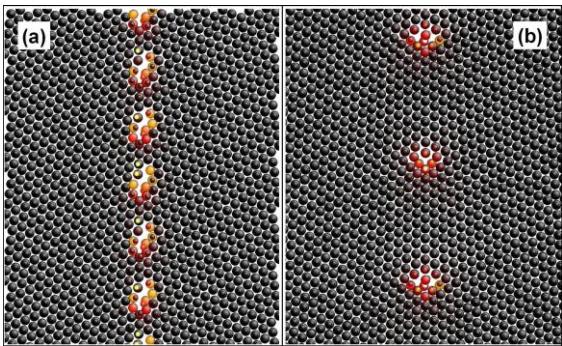
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Homogénéisation numérique : Applications

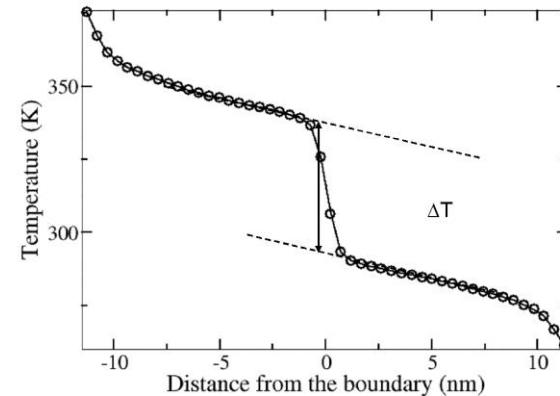
Effect of grain boundaries on the thermal conductivity of SiC polycrystals from discrete to continuous modeling

Jour. Applied Physics, Crocombe J.P., Gélébart L., 106, 083520 (2009)

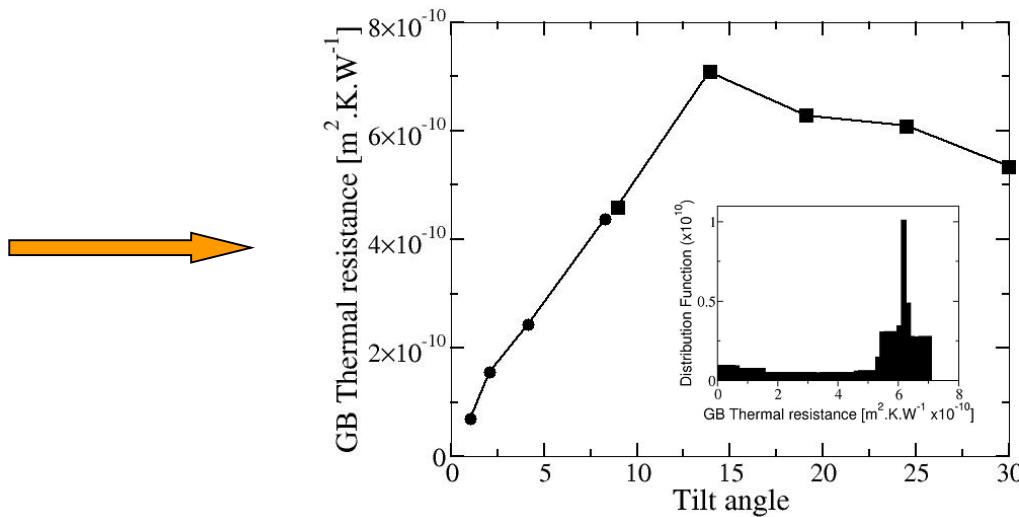
Grain boundary definition



Thermal calculation



$$q = -h\Delta T$$





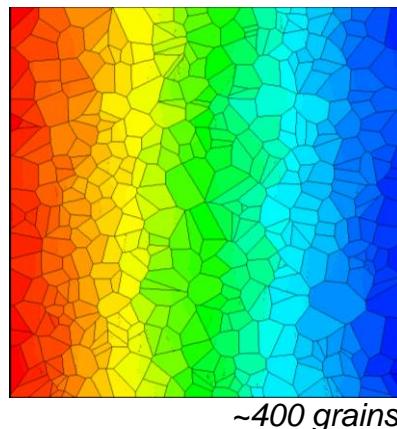
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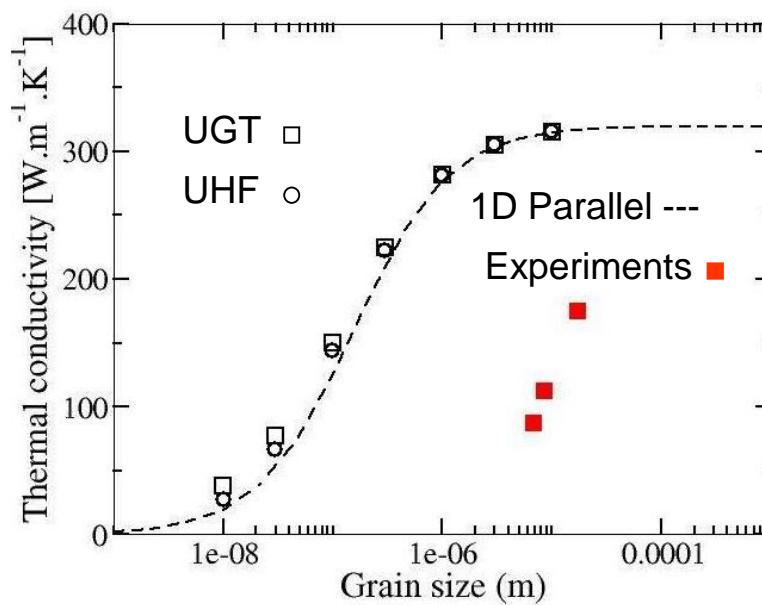
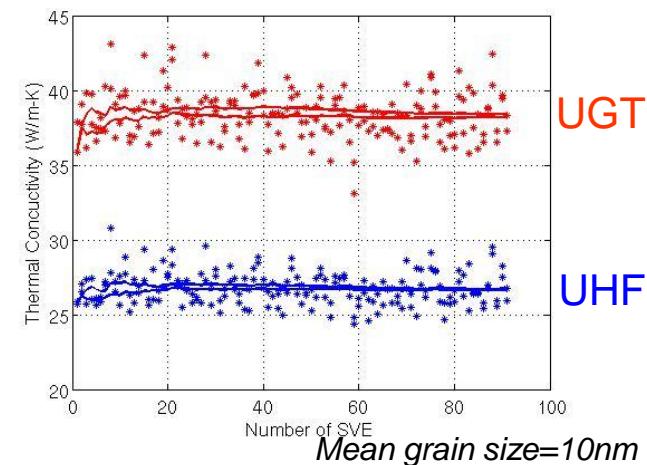
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Heterogeneities :
Grain sizes
GB resistances

Finite Element simulation
UGT or UHF boundary conditions



Effective (mean) behaviour



Maximum relative difference (EF-1Dparallel)
-> less than 20%

Large discrepancy with experimental results
-> too perfect grain boundaries in MD



Homogénéisation numérique : principes

Définition d'un chargement :

$$\left\{ \begin{array}{l} \operatorname{div}(\underline{\underline{\sigma}}) = 0 \\ \underline{\underline{\sigma}} = K : \underline{\underline{\varepsilon}} \\ (\underline{u}, \underline{\underline{\sigma}} \cdot \underline{n})_{\partial\Omega} \in E_{X_0} \quad \longrightarrow \quad \text{Chargement moyen} \\ (\underline{u}, \underline{\underline{\sigma}} \cdot \underline{n})_{\partial\Omega} \in E^{CL} \quad \longrightarrow \quad \text{Type de CL} \end{array} \right.$$

Exemples de chargements moyens

$$E_{\Sigma_0} = \left\{ (\underline{v}, \underline{t})_{\partial\Omega} \quad / \quad \frac{1}{V} \int_{\partial\Omega} \underline{t} \otimes^s \underline{x} ds = \underline{\underline{\Sigma_0}} \right\} \quad \longrightarrow \quad \text{Contrainte moyenne} = \underline{\underline{\Sigma_0}}$$

$$E_{E_0} = \left\{ (\underline{v}, \underline{t})_{\partial\Omega} \quad / \quad \frac{1}{V} \int_{\partial\Omega} \underline{v} \otimes^s \underline{n} ds = \underline{\underline{E_0}} \right\} \quad \longrightarrow \quad \text{Déformation moyenne} = \underline{\underline{E_0}}$$



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Homogénéisation numérique : principes

Définition du comportement élastique apparent

A partir de 6 chargement linéairement indépendants*
(* pas de relation linéaire entre les 6 contraintes moyennes)

Approche mécanique

$$\overline{\underline{\sigma}^I} = \tilde{K}^{mech} : \overline{\underline{\varepsilon}^I}$$

Approche énergétique

$$\overline{e^{IJ}} = \overline{\underline{\sigma}^I : \underline{\varepsilon}^J} = \frac{1}{2} \overline{\underline{\varepsilon}^I} : \tilde{K}^{ener} : \overline{\underline{\varepsilon}^J}$$

Equivalence approche énergétique / approche mécanique



Tout chargement d'un « jeu de CL » doit respecter la condition de Hill

$$\frac{1}{2} \overline{\underline{\sigma} : \underline{\varepsilon}} = \frac{1}{2} \overline{\underline{\sigma}} : \overline{\underline{\varepsilon}} \quad \longleftrightarrow \quad \int_{\partial\Omega} (\underline{t} - \overline{\underline{\sigma}} \cdot \underline{n}) \cdot (\underline{u} - \overline{\underline{\varepsilon}} \cdot \underline{x}) dS = 0$$



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Homogénéisation numérique : principes

Les 3 « jeux de CL » classiques satisfaisant Hill

CL en Déformation Homogène au contour

$$E^{DH} = \left\{ (\underline{v}, \underline{t})_{\partial\Omega} \middle/ \underline{v} = \underline{\underline{\varepsilon}} \cdot \underline{x} \quad \text{avec} \quad \underline{\underline{\varepsilon}} = \frac{1}{V} \int_{\partial\Omega} \underline{v} \otimes^s \underline{n} ds \right\}$$

CL en Contrainte Homogène au contour

$$E^{CH} = \left\{ (\underline{v}, \underline{t})_{\partial\Omega} \middle/ \underline{t} = \underline{\underline{\sigma}} \cdot \underline{n} \quad \text{avec} \quad \underline{\underline{\sigma}} = \frac{1}{V} \int_{\partial\Omega} \underline{t} \otimes^s \underline{x} ds \right\}$$

CL Périodiques

$$E^P = \left\{ \begin{array}{l} (\underline{v}, \underline{t})_{\partial\Omega} \\ \underline{v}(\underline{x} + \underline{h}) = \underline{v}(\underline{x}) + \underline{\underline{\varepsilon}} \cdot \underline{h} \quad \text{et} \quad \underline{t}(\underline{x} + \underline{h}) = -\underline{t}(\underline{x}) \end{array} \quad \text{avec} \quad \underline{\underline{\varepsilon}} = \frac{1}{V} \int_{\partial\Omega} \underline{v} \otimes^s \underline{n} ds \right\}$$